# Study of the Vibronic Transitions of Gd<sup>3+</sup> and Eu<sup>3+</sup> in Crystalline Materials and Glasses of the Same Composition

A. Ellens,<sup>1</sup> B. Salemink, A. Meijerink, and G. Blasse

Debye Institute, Utrecht University, Department of Condensed Matter, P.O. Box 80000, NL-3508 TA Utrecht, The Netherlands

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The electron-phonon coupling strength of Gd<sup>3+</sup> and Eu<sup>3+</sup> ions in crystalline and glass phases of the same chemical composition are studied by measuring the vibronic transition probabilities of the 4f<sup>n</sup> transitions of these ions. The experimentally derived vibronic transition probabilities are very small. Within the experimental error, the electron-phonon coupling strength of the lanthanide ions in the different phases is the same. © 1998 Academic Press

## 1. INTRODUCTION

The interaction of the 4f electrons of lanthanide ions with the lattice vibrations (electron–phonon coupling) is weak but has a significant influence on the spectroscopy of lanthanide ions. Knowledge of the electron–phonon coupling is therefore not only interesting from a fundamental point of view, but also important for the research on the application of systems containing lanthanide ions, like lasers, scintillators, and phosphors (1, 2).

Up to now, vibronic transitions and vibronic transition probabilities ( $A_{vib}$ ) were mainly measured in crystalline host materials. In this study we compare the vibronic transition probabilities of lanthanide ions in crystalline materials and glasses of the same chemical composition. On the basis of the less dense structure of glasses, the ion–ligand distances will be larger, and hence one might expect weaker electron– phonon coupling of the lanthanide 4f electrons with the lattice vibrations in the glasses than in the crystalline material. We present the results on the vibronic transition probabilities of Gd<sup>3+</sup> and Eu<sup>3+</sup> as dopants in the crystalline and the glass modifications of LaB<sub>3</sub>O<sub>6</sub> and LiLaP<sub>4</sub>O<sub>12</sub>. The reason that we have chosen to investigate the ions Gd<sup>3+</sup> and Eu<sup>3+</sup> is that vibronic transitions of these ions have been investigated thoroughly in other systems (3–5). The  ${}^{6}P_{7/2} \Rightarrow {}^{8}S_{7/2}$  transition of Gd<sup>3+</sup> was measured because this transition is an excellent one to study the vibronic interaction of the ion with its surroundings (3, 6). For Eu<sup>3+</sup> the  ${}^{5}D_{0,2} \leftarrow {}^{7}F_{0}$  transitions were studied.

 $LaB_3O_6$  and  $LiLaP_4O_{12}$  have been chosen as hosts since these compounds show vibrational frequencies at high energies, so that the corresponding vibronic features are well separated from the inhomogeneously broadened zerophonon line in the glass. Besides, these materials are known to have the same chemical composition and the same shortrange ordering in the crystalline material as in the glass (7–9).

### 2. EXPERIMENTAL

Crystalline LaB<sub>3</sub>O<sub>6</sub> and LiLaP<sub>4</sub>O<sub>12</sub> doped with Gd<sup>3+</sup> or Eu<sup>3+</sup> were prepared according to way described in Refs. (10) and (11), respectively. To obtain glasses, the starting materials were heated at 1200°C for half an hour in quartz crucibles and subsequently poured out in carbon crucibles (at room temperature).

The crystalline samples were checked by X-ray powder diffraction analysis, and were found to be single phase. The glass samples did not contain crystalline material, as far it could be detected by X-ray powder diffraction. Transmission spectroscopy showed that the crystalline and the glass samples did not contain optical impurities.

Low- and high-resolution excitation and emission measurements and decay time measurements were performed using a SPEX DM3000F spectrofluorometer, an excimerlaser pumped-dye laser, and a Nd: YAG laser setup, described elsewhere (6).

#### **3. STRUCTURE**

The structure of crystalline LaB<sub>3</sub>O<sub>6</sub> is based on the boronoxygen units  $[B_6O_{12}]_{\infty}^{6-}$  which consist of two BO<sub>4</sub> tetrahedra and four BO<sub>3</sub> triangles linked to each other and forming long chains. Parallel chains of these boron-oxygen units are joined together by lanthanide ions, situated

<sup>&</sup>lt;sup>1</sup>To whom correspondence should be addressed. Present address: Institut für anorganische, analytische und physikalische Chemie, Freiestrasse 3, CH-3000 Bern 9, Switzerland. Phone: +41316314248. Fax: +4131 6313993. E-mail: ARD@iac.unibe.ch.

between the chains. The lanthanide ion is at the center of a distorted ten-cornered polyhedron of oxygen ions. The unit cell of crystalline LaB<sub>3</sub>O<sub>6</sub> is monoclinic with space group I2/c (10, 12).

The unit cell of crystalline LiLaP<sub>4</sub>O<sub>12</sub> is monoclinic with space group C2/c. The basic structural units are helical ribbons (PO<sub>3</sub>)<sub>n</sub> formed by corner-sharing PO<sub>4</sub> tetrahedra. The lanthanide ion is at the center of a LaO<sub>8</sub> dodecahedron (13).

#### 4. RESULTS

#### Gd<sup>3+</sup>

In Figs. 1a and 1b, the low-resolution emission spectra of the  ${}^{6}P_{7/2} \Rightarrow {}^{8}S_{7/2}$  transition in glass and crystalline LaB<sub>3</sub>O<sub>6</sub>: Gd<sup>3+</sup> at 4.2 K are presented as examples. At about 32,000 cm<sup>-1</sup> the zero-phonon line is found, and at lower energies (around 30,500 cm<sup>-1</sup>) vibronic transitions are observed. These vibronic transitions are due to coupling with the high-energetic vibrations of the host lattice. In the crystalline material (Fig. 1b) vibronic transitions are also found at lower energies, but in the glass phase (Fig. 1a) these vibronics cannot be separated from the inhomogeneously broadened zero-phonon line.

The high-energetic vibronics, which can clearly be distinguished from the zero-phonon line, are in the crystalline phase at a distance of  $1250 \text{ cm}^{-1}$  from the zero-phonon line (due to vibrations of the BO<sub>4</sub> group) and at 1410 cm<sup>-1</sup> (due to B–O stretching vibrations of the BO<sub>3</sub> group). The



**FIG.1.** Emission spectrum in the  ${}^{6}P_{7/2} \Rightarrow {}^{8}S_{7/2}$  region of Gd<sup>3+</sup> in (a) glass and (b) crystalline LaB<sub>3</sub>O<sub>6</sub>, both at 4.2 K. Excitation is in the  ${}^{6}I_{7/2}$  level of Gd<sup>3+</sup>. The vibronic regions are magnified 50 ×.

positions of these vibronic transitions are in agreement with the data from the literature (10). The fact that the energetic position of the vibronic transitions are within the experimental error the same in the glass and the crystalline material is in agreement with the results from Refs. (7–9) that these metaborate glasses contain the same borate groups which are found in the crystalline modification.

In the emission spectra of  $LiLaP_4O_{12}:Gd^{3+}$  glass and crystalline powder, not shown here, the high-energetic vibronics are found at about 1090 and 1220 cm<sup>-1</sup>. The position of these vibronic transitions is in agreement with data from the literature on Raman and IR spectroscopy on these materials. These vibronic transitions are ascribed to coupling with P–O stretching vibrations (9–11).

From the decay times and the *R* values (defined as the ratio of the integrated vibronic intensity and the integrated zero-phonon line intensity), the transition probabilities for the zero-phonon line  $(A_{zp})$  and the vibronics  $(A_{vib})$  of the  ${}^{6}P_{7/2} \Rightarrow {}^{8}S_{7/2}$  transitions of Gd<sup>3+</sup> can be calculated, as has been described for other systems (3, 6, 14). The results for the Gd<sup>3+</sup> containing systems are given in Table 1.

It should be realized that for the R value and hence the vibronic transition probabilities, only the high-energetic vibronics have been taken into account. The total error in the vibronic transition probabilities is about 50%, mainly due to the fact that they are so weak relative to the zero-phonon line.

The *R* value obtained for crystalline LaB<sub>3</sub>O<sub>6</sub>: Gd<sup>3+</sup> (0.026) differs from the result presented by Verweij *et al.* (10) on GdB<sub>3</sub>O<sub>6</sub> (0.12). This difference can be explained by the fact that Verweij *et al.* used a 100% Gd<sup>3+</sup> crystalline material in which reabsorption may occur. The reabsorbed light is then redistributed over the zero-phonon and vibronic lines, and so the intensity of the vibronic sideband is enhanced with respect to the zero-phonon line (15). To obtain a correct *R* value, measurements should be performed on a diluted Gd<sup>3+</sup> sample.

Eu<sup>3+</sup>

For  $Eu^{3+}$  in the glass and crystalline modifications of  $LaB_3O_6$  and  $LiLaP_4O_{12}$ , two excitation transitions were

TABLE 1R Values, Decay Times, and Transition Probabilities of  $Gd^{3+}$ in Crystalline and Glassy LaB<sub>3</sub>O<sub>6</sub> and LiLaP<sub>4</sub>O<sub>12</sub> at 4.2 K

System	$^{6}P_{7/2} \Rightarrow ^{8}S_{7/2}$	Decay (μs) <sup>6</sup> P <sub>7/2</sub>	${}^{6}\mathrm{P_{7/2}} \Rightarrow {}^{8}\mathrm{S_{7/2}}$	$\begin{array}{c} A_{\mathrm{vib}} \left(\mathrm{s}^{-1}\right) \\ {}^{6}\mathrm{P}_{7/2} \! \Rightarrow  {}^{8}\mathrm{S}_{7/2} \end{array}$
$LaB_{2}O_{\epsilon}:Gd^{3+}$				
Crystal	0.026	4000	244	7
Glass	0.02	3400	288	6
$LiLaP_4O_{12}:Gd^{3+}$				
Crystal	0.035	6570	147	5
Glass	0.02	4550	215	4

studied: the  ${}^{5}D_{0} \leftarrow {}^{7}F_{0}$  and the  ${}^{5}D_{2} \leftarrow {}^{7}F_{0}$  transitions. These transitions were chosen because the vibronic sidebands in the spectra of these transitions were not obscured by the presence of the zero-phonon lines of other transitions. Due to the very low oscillator strength of the  ${}^{5}D_{0} \leftarrow$  ${}^{7}F_{0}$  transition, the transitions on Eu<sup>3+</sup> were studied with the laser setups.

Here we show only the spectra of the  ${}^{5}D_{0} \leftarrow {}^{7}F_{0}$ transition of Eu<sup>3+</sup> in glass and crystalline LaB<sub>3</sub>O<sub>6</sub>, see Figs. 2a and 2b, respectively. The spectra are corrected for the dye laser intensity. The zero-phonon line is found at some 17,300 cm<sup>-1</sup>, and vibronic transitions, due to coupling with high-energy vibrations, are found at a distance of some 1100 and 1400 cm<sup>-1</sup>. In the crystalline materials (Fig. 2b) low-energetic vibronics are also found, which in the glass (Fig. 2a) cannot be separated from the zero-phonon line. From the spectra it is clear that the relative intensity (*R*) of the vibronic transitions in the crystal is much larger than in the glass.

The *R* values and the vibronic transition probabilities, calculated for the  ${}^{5}D_{0} \leftarrow {}^{7}F_{0}$  and the  ${}^{5}D_{2} \leftarrow {}^{7}F_{0}$  transitions, are given in Table 2. The vibronic transition probability of the  ${}^{5}D_{0} \leftarrow {}^{7}F_{0}$  transition is derived from the decay time of the  ${}^{5}D_{0}$  level and the relative emission intensity of the  ${}^{5}D_{0} \Rightarrow {}^{7}F_{0}$  transition in the total  ${}^{5}D_{0}$  emission spectrum (branching ratio). From  $A_{zp}$  of the  ${}^{5}D_{0} \Rightarrow {}^{7}F_{0}$  transition, which is the same in emission as in excitation and the

10 а 8 50 X Relative Intensity (a.u.) 2 10 0 b 10 X 8 6 4 2 0 16500 17500 18500 19500 Energy (cm<sup>-1</sup>)

**FIG.2.** Excitation spectrum in the  ${}^{5}D_{0} \leftarrow {}^{7}F_{0}$  region of Eu<sup>3+</sup> in (a) glass and (b) crystalline LaB<sub>3</sub>O<sub>6</sub>, both at 4.2 K. The emission of the  ${}^{5}D_{0} \Rightarrow {}^{7}F_{2}$  transition is monitored. The vibronic regions in the figures are magnified 50 × for the glass and 10 × for the crystalline material.

TABLE 2*R* Values, Decay Times, and Transition Probabilities of Eu<sup>3+</sup>in Crystalline and Glass LaB<sub>3</sub>O<sub>6</sub> and LiLaP<sub>4</sub>O<sub>12</sub> at 4.2 K ofTtransitions <sup>5</sup>D<sub>0</sub>  $\Leftarrow$  <sup>7</sup>F<sub>0</sub> and <sup>5</sup>D<sub>2</sub>  $\Leftarrow$  <sup>7</sup>F<sub>0</sub>

System	<i>R</i> values ${}^{5}D_{0} \Leftarrow {}^{7}F_{0}/$ ${}^{5}D_{2} \Leftarrow {}^{7}F_{0}$	Decay ( $\mu$ s) ${}^{5}D_{0}$	$\begin{array}{c} A_{zp} \left( \mathbf{s}^{-1} \right) \\ {}^5\mathrm{D_0} \Leftarrow {}^7\mathrm{F_0} / \\ {}^5\mathrm{D_2} \Leftarrow {}^7\mathrm{F_0} \end{array}$	$\begin{array}{c} A_{\mathrm{vib}}  (\mathrm{s}^{-1}) \\ {}^{5}\mathrm{D}_{\mathrm{0}} \Leftarrow {}^{7}\mathrm{F}_{\mathrm{0}} / \\ {}^{5}\mathrm{D}_{\mathrm{2}} \Leftarrow {}^{7}\mathrm{F}_{\mathrm{0}} \end{array}$
$LaB_3O_6:Eu^{3+}$				
Crystal	0.51/0.10	3366	1.5/35	1/3
Glass	0.02/0.01	1356	8/200	0.2/2
LiLaP <sub>4</sub> O <sub>12</sub> :Eu <sup>3</sup>	+			
Crystal	0.21/0.18	4500	0.3/14	0.1/3
Glass	0.03/0.01	3500	2/175	0.03/2

*R* value, the vibronic transition probability of the  ${}^{5}D_{0} \leftarrow {}^{7}F_{0}$  transition can be calculated.

The transition probability of the  ${}^{5}D_{2} \leftarrow {}^{7}F_{0}$  transition is calculated by relating the intensities of the zero-phonon lines of the two transitions studied ( ${}^{5}D_{0} \leftarrow {}^{7}F_{0}$  and the  ${}^{5}D_{2} \leftarrow {}^{7}F_{0}$ ) in the excitation spectra. Hence, from the transition probability derived for the  ${}^{5}D_{2} \leftarrow {}^{7}F_{0}$  transition and the *R* value, the vibronic transition probability for this transition can be derived. Because the branching ratio of the  ${}^{5}D_{0} \Rightarrow {}^{7}F_{0}$  transition is very small, the procedure to derive the vibronic transition probabilities via the branching ratio of the  ${}^{5}D_{0} \Rightarrow {}^{7}F_{0}$  transition introduces a large error (~100%) in the values of the vibronic transition probabilities for Eu<sup>3+</sup>.

#### 5. DISCUSSION AND CONCLUSIONS

The data presented in Tables 1 and 2 show that the transition probabilities of the zero-phonon lines differ significantly between the crystalline materials and the glasses, especially for  $Eu^{3+}$ . The larger values for the transition probabilities of the zero-phonon lines in the glass can be due to the lower symmetric environment of the lanthanide ion in the glasses. Since the transitions on  $Eu^{3+}$  are more sensitive to the environment (4), lowering of symmetry will have a larger influence on the transition probabilities of  $Eu^{3+}$ .

For a good comparison of vibronic intensities of lanthanide ions in different systems, not the R value but the vibronic transition probabilities must be compared (6), because the R value is dependent on the value of the zero-phonon line transition probability.

Vibronic transition probabilities scale with the ion– ligand distance  $R_{\rm L}^{-6}$  (static coupling) or  $R_{\rm L}^{-10}$  (dynamic coupling), and hence (14) smaller vibronic intensities are expected for the glasses, which are less dense. Since the differences in density between the crystalline and the glasses are small, a small effect is expected. It is clear from the tables that the differences in the vibronic transition probabilities are smaller than the experimental errors, which are estimated to be some 50% for the values of  $Gd^{3+}$  and more than 100% for  $Eu^{3+}$ . These large errors are mainly due to the very low intensities of the vibronic transitions, and the very small  $A_{zp}$  values for  $Eu^{3+}$ . This means that the differences in vibronic transition probabilities between the crystalline material and the glass are too small to be estimated from these systems and with this method. Possibly in other systems with a stronger electron-phonon coupling than  $Gd^{3+}$  and  $Eu^{3+}$  (see for example Refs. (6, 16–18)) and more intense vibronic transitions, the vibronic transition probabilities can be determined with higher accuracy. This is required if there is a difference in the electron-phonon coupling strength between the glass and crystalline material.

Hence, within the experimental error, the electronphonon coupling of the lanthanide ions  $Gd^{3+}$  and  $Eu^{3+}$  are the same in the glass and the crystalline phases of the studied host lattices.

#### REFERENCES

 G. Blasse and B. C. Grabmaier, "Luminescent Materials." Springer-Verlag, Berlin, 1994.

- 2. B. Henderson and G. F. Imbusch, "Optical Spectroscopy of Inorganic Solids." Oxford Univ. Press, Oxford, 1989.
- J. Sytsma, W. van Schaik, and G. Blasse, J. Phys. Chem. Solids 52, 419 (1991).
- 4. G. Blasse, Inorg. Chim. Acta 167, 33 (1990).
- 5. G. Blasse, Int. Rev. Phys. Chem. 11, 71 (1992).
- 6. A. Ellens, S. Schenker, A. Meijerink, and G. Blasse, J. Lumin. 69, 1 (1996).
- I. N. Chakroborty, J. E. Shelby, and R. A. Condrate, Sr., J. Am. Ceram. Soc. 67, 782 (1984).
- I. N. Chakroborty, D. E. Day, G. H. Schoenmakers, J. C. Lapp, and J. E. Shelby, J. Am. Ceram. Soc. 68, 368 (1985).
- 9. D. W. Hall, S. A. Brawer, and M. J. Weber, Phys. Rev. B 25, 2828 (1982).
- J. W. M. Verweij, G. F. Imbusch, and G. Blasse, J. Phys. Chem. Solids 50, 813 (1989).
- 11. J. W. M. Verwey and G. Blasse, Chem. Mater. 2, 91 (1990).
- G. K. Abdullaev, Kh. S. Mamedov, and G. G. Dzhafarof, Krisallografiya 20, 265 (1975).
- 13. H. Y. P. Hong, Mater. Res. Bull. 10, 635 (1975).
- C. de Mello Donega, A. Meijerink, and G. Blasse, J. Phys.: Condens. Matter 4, 8889 (1992).
- C. de Mello Donega, A. Meijerink, and G. Blasse, J. Lumin. 62, 189 (1994).
- A. Ellens, H. Andres, M. L. H. ter Heerdt, R. T. Wegh, A. Meijerink, and G. Blasse, *J. Lumin.* 66/67, 240–243 (1996).
- A. Ellens, H. Andres, A. Meijerink, and G. Blasse, *Phys. Rev. B* 55, 173 (1997).
- A. Ellens, H. Andres, M. L. H. ter Heerdt, R. T. Wegh, A. Meijerink, and G. Blasse, *Phys. Rev. B* 55, 180 (1997).